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Research Project Status Update: Final Report (Stage Gate Grant Year 2)

Date: (Originally Submitted) <u>10/15/2024</u>

University / College: Lehigh University

Project Title: Numerical Modeling of Tooling Improvements for Forging Processes

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Please complete the below questions to report your project status. The answers will be shared with the FIA Technical Committee.

1. Progress on accomplishments vs. deliverables:

- Printing and evaluation of acceptably dense material coupons and their corresponding cylindrical Gleeble samples
 - o INC625 (pure)
 - INC625 + 5w%TiC
- Stress-strain curves obtained via Gleeble-3500 thermo-mechanical compression experiments for following materials (a total of 35 successful tests were performed):
 - H13 (annealed)
 - H13 (hardened [quenched and tempered])
 - o INC625 (pure)
 - INC625 + 5w%TiC
- Use of stress-strain curves to generate material models and resulting simulations in DEFORM-2D

Summary of the above points:

This project, "Numerical Modeling of Tooling Improvements for Forging Processes" builds off of research conducted for the FIERF Stage Gate Previous Year Grant project, "Improvements of H13 forging tool life through TiC-reinforced Inconel 625 coatings produced by additive manufacturing".

The first stage gate of that project involved the use of Lehigh University's Renishaw AM400 SLM machine to iteratively print coupons of the materials being investigated: pure Inconel 625 (INC625) and INC625 reinforced with varying percentages (by weight) of Titanium Carbide (TiC). This combination of the INC625 nickel-based superalloy with the hard, refractory ceramic TiC forms a Metal Matrix Composite (MMC).

To obtain coupons of sufficient density, a sliding window approach was taken to adjust the controllable parameters that exhibit the most significant impact on SLM printing: laser power, hatch space, exposure time, and layer thickness. The sliding window approach involved selecting two variables being chosen to create a matrix while keeping the remaining variables at a fixed value. Once the first print is completed, the results are evaluated; it can be observed which parameters should be altered, and they are changed accordingly.

After reasonable density was achieved by visual inspection of the coupons, the test coupons (which are printed in cube form) were cut into rows and mounted in epoxy, then ground and polished in preparation for metallography evaluation.



Figure 1: (Left) Pure Inconel 625 printed coupons, (Middle) INC 625 + 5w%TiC printed coupons, (Right) INC625 + 5w%TiC printed coupons (desired density not achieved)

The metallography process is comprised of grinding, polishing, and etching—with grinding and polishing being required to assess the coupons' density percentage via Light Optical Microscopy (LOM), and etching is necessary to expose the microstructure and metal flow.

While several iterations of prints were performed for the INC625 + 10w%TiC, acceptable density was not able to be achieved; instead, the focus turned to optimizing the INC625 + 5w%TiC, which was able to achieve a density above 99.5%. The increase of TiC in the INC625 + 10w%TiC markedly altered the flow characteristic during printing; this, in conjunction with the differing morphologies of the powders (the INC625 powder being spherical while the TiC powder having a flakier shape) likely had a significant effect on the printability of the resulting mixed powder.

The predominant goal for this stage gate of the project was the prediction of the behavior of the material to produce process simulations by using the DEFORM finite element analysis (FEA) software package—specifically, DEFORM-2D. While DEFORM does contain materials libraries that can be used for

process simulations, for materials for which libraries may not exist—such as specific MMCs—users can provide the resulting stress-strain curves from thermo-mechanical experiments to develop material models that can be used for process simulations.

Once the cylindrical samples were printed atop the sacrificial H13 cylinders, they were machined to the appropriate size and cut off from the H13 base cylinders. At this point, they are ready for use with the Gleeble 3500 to perform hot compression tests. These Gleeble experiments used multipart ISO-T anvils or single-piece tungsten carbide anvils. The cylindrical Gleeble sample then had 2 Type K thermocouple wires welded and placed between the anvils with a holding force of 0.2 to 0.4 kN.

The test begins by heating the sample with resistive heating; as current passes through the sample and the sample itself expands, the moving ram adjusts the position of the anvils to keep the same holding force throughout the heating process. After the user-specified temperature is reached, the unit maintains that temperature for a period (dwelling) to ensure a uniform temperature distribution throughout the sample. Next, the ram compresses the sample at the user-specified strain rate. Over the entirety of the process, the temperature, stress, strain, force, and ram position are all recorded into a file converted into a Microsoft Excel CSV to generate stress-strain plots. These plots—generated at different user-specified temperatures and strain rates—perform the material characterization in DEFORM.



Figure 2: (Left) A sample mounted between the Gleeble 3500 anvils; (Right) A heated sample undergoing testing in the Gleeble 3500.

Gleeble tests were performed with annealed (as-received) H13, hardened H13 (quenched and tempered following the standard heat treatment), pure INC625, and INC625 + 5w%TiC. A total of 35 successful tests have been performed to obtain data for material modeling within DEFORM-2D; *Table 1*, below, outlines the test breakdown:

Material	Experiment	Experiment Strain	Number of Successful
	Temperatures	Rates	Tests
H13 (annealed)	600°C 650°C 700°C	15 ⁻¹ 55 ⁻¹ 105 ⁻¹ 205 ⁻¹	17
H13 (hardened)	650°C 750°C	1s ⁻¹ 10s ⁻¹	4
INC625 (pure)	750°C 800°C 850°C	1s ⁻¹ 5s ⁻¹ 10s ⁻¹ 20s ⁻¹	6
INC625 + 5w%TiC	750°C 800°C 825°C 850°C	15 ⁻¹ 55 ⁻¹ 105 ⁻¹ 205 ⁻¹	8
	tal Successful Tests:	35	

Table 1: Summary of successful Gleeble tests for materials characterization.

The 4 materials above were tested at the listed temperatures and strain rates—H13 data was gathered to establish a baseline performance for forging die material, while the pure INC625 and INC625 + 5w%TiC Gleeble tests were completed to observe the performance difference between pure INC625 and INC625 and INC625 with 5w%TiC reinforcement.

Throughout these Gleeble experiments, certain temperatures were found to be unconducive for testing at certain temperature ranges—specifically, the initial testing temperatures were found to be too low. It was established that the Inconel 625—regardless of whether it was pure or contained the TiC reinforcement—was too hard to be tested below 750°C, with the Gleeble not providing accurate results at any temperature below 800°C. The reluctance for INC625 to interact with the lower temperature is to be expected, as the material has excellent temperature resistance (a condition considered in this project's initial materials selection phase). Due to this characteristic of INC625, the samples were tested on the Gleeble at higher temperatures—775°C, 800°C, and 850°C—using the strain rates of 1s⁻¹ and 10s⁻¹. These test conditions allow for better estimation of material behavior at lower temperatures when generating the DEFORM simulations.



Figure 3: Stress-strain curves of INC625 and INC625 + 5w%TiC Gleeble experiment results.

Once a sufficient number of tests were performed for each material, the process of importing the stress-strain curves for use in DEFORM took place. A summary of DEFORM simulations—informed by the data collected over the course of the successful Gleeble experiments—follows.

Figure 4 shows simulation excerpts taken at 100 and 175 steps, from a total of 200 steps. The material sample's pre-deformation dimensions were 15mm height by 10mm diameter; total displacement specified for each compression test was 10mm. The range of temperatures tested were between 650°C and 800°C for the as-received H13, as well as for the hardened H13. Strain rates tested for all materials were 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹.



Figure 4: Example of comparison of H13 annealed (left) and hardened (right) at step 100 and step 175 (out of 200) of their respective simulations.

Evaluation of the local strain, stress, and strain for both the annealed and hardened H13 are presented; a listing of all parameters used for the DEFORM 2D simulations can be found in *Appendix B*. Taking into account these parameters, the materials were tested at different ranges of temperatures: H13 samples—both annealed and hardened—underwent compression tests in the temperature range of 650°C and 800°C, while the pure INC625 samples and the MMC samples underwent compression tests in the range 750°C and 850°C. The points of samples that underwent extreme temperature and strain are shown in *Figure 5*. In all cases, the range of strain rates of 1s⁻¹ through 20s⁻¹ was considered to evaluate the center and border regions, as indicated in *Figure 5*.



Figure 5: Regions of evaluation (center and border) with 2D and 3D views.

Selection of the appropriate higher temperatures and strain rates (different speeds) for Gleeble compression tests yielded curves for effective process modeling in DEFORM-2D. Shown below are the stress-strain curves obtained through heat and compression Gleeble experiments performed with the cylindrical, annealed H13 samples. These experiments with the annealed H13 samples were performed at the temperatures of 650°C (refer to *Figure 6*) and 800°C (refer to *Figure 7*) and at strain rates of 1s⁻¹, $5s^{-1}$, $10s^{-1}$, and $20s^{-1}$, with the corresponding results shown below:



Figure 6: Plots of stress-strain curves obtained from Gleeble hot compression tests performed on H13 (annealed) samples at 650°C and strain rates of 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹.



Figure 7: Plots of stress-strain curves obtained from Gleeble hot compression tests performed on H13 (annealed) samples at 800°C and strain rates of 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹.

Ultimately, it was observed that the DEFORM-2D simulations closely matched the results obtained through the physical experiments performed on the Gleeble. *Figure 8* and *Figure 9* are the DEFORM-2D simulations of the annealed H13's local behavior of the effective strain (in mm/mm) at steps 100 and 175 (out of 200 steps total). *Figure 8* shows this simulation performed at 650°C and strain rates of 1s-1 and 20s-1; *Figure 9* shows this simulation performed at 800°C at strain rates of 1s⁻¹ and 20s⁻¹.



Figure 8: DEFORM-2D simulation results of annealed H13's effective strain behavior for steps 100 and 175 (out of 200) at temperature of 650°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 9: DEFORM-2D simulation results of annealed H13's effective strain behavior for steps 100 and 175 (out of 200) at temperature of 800°C and strain rates of 1s⁻¹ and 20s⁻¹.

Table 2, presented below, summarizes the effective strain (in mm/mm) observed at steps 100 and 175 for the center and border regions of the annealed H13; these simulations' effective strains are shown at temperatures 650°C, 700°C, 750°C, and 800°C and strain rates 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹. It can be observed that as the speed—dictated by strain rate—is increased, the strain itself decreases slightly; almost no variation is seen as the temperature is increased, save for the experiments run at 800°C, where a decrease occurs:

Table 2: Overview of effective strain (designated as "Strain-Effective (mm/mm)" in DEFORM-2D) from annealed H13 simulations in the range of temperatures from 650°C to 800°C and strain rates from 1s⁻¹ to 20s⁻¹.

Strain Ra	te	1s ⁻¹ Center	1s ⁻¹ Border	5s⁻¹ Center	5s⁻¹ Border	10s⁻¹ Center	10s⁻¹ Border	20s ^{−1} Center	20s⁻¹ Border
Tempe	erature								
T=	Step 100	0.6	0.45	0.59	0.45	0.59	0.45	0.59	0.45
650 C	Step 175	1.33	0.77	1.33	0.77	1.33	0.77	1.33	0.77
T=	Step 100	0.6	0.45	0.59	0.45	0.59	0.45	0.59	0.45
700 C	Step 175	1.33	0.77	1.33	0.77	1.33	0.77	1.33	0.77

T=	Step 100	0.6	0.46	0.66	0.45	0.59	0.45	0.59	0.45
750 C	Step 175	1.19	0.77	1.33	0.77	1.33	0.77	1.33	0.77
T=	Step 100	0.53	0.46	0.66	0.45	0.52	0.45	0.59	0.45
800 C	Step 175	1.19	0.77	1.19	0.77	1.19	0.77	1.19	0.77

Below, DEFORM-2D simulation excerpts *Figure 10* and *Figure 11* show the local behavior at steps 100 and 175 (out of 200) of the effective stress (in MPa) of the annealed H13 undergoing temperatures at 650°C and 800°C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 10: DEFORM-2D simulations showing the stress behavior—at steps 100 and 175—of annealed H13 at a temperature of 650°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 11: DEFORM-2D simulations showing the stress behavior—at steps 100 and 175—of annealed H13 at a temperature of 800°C and strain rates of 1s⁻¹ and 20s⁻¹.

As outlined in **Table 3**, within steps 100 and 175 the stress from the center to the border decreases the increasing temperature and increases with the higher speeds—that is, strain rates—of the process. It is also evident that between steps 100 and 175, stress decreases.

Table 3: Overview of effective strain (designated as "Strain-Effective (MPa)" in DEFORM-2D) from annealed H13 simulations in the range of temperatures from 650° C to 800° C and strain rates from $1s^{-1}$ to $20s^{-1}$.

Strain Ra	tei erature	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
T=	Step 100	222	183	302	252	345	289	395	332
650 C	Step 175	205	185	283	258	325	297	373	311
T=	Step 100	183	154	277	227	317	251	383	301
700 C	Step 175	185	157	258	233	297	269	342	311
T=	Step 100	164	144	227	201	261	233	332	270
750 C	Step 175	147	127	233	183	269	214	311	250
T=	Step 100	144	124	201	176	233	205	301	238
800 C	Step 175	127	108	183	158	241	214	281	220

The DEFORM-2D simulation excerpts in *Figure 12* and *Figure 13* show the local effective strain rate (in [mm/mm]/s) behavior at steps 100 and 175 (out of 200) of the annealed H13, at temperatures 650°C (*Figure 12*) and 800°C (*Figure 13*) and at strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 12: DEFORM-2D simulations showing the effective strain rate ([mm/mm]/s) behavior—at steps 100 and 175—of annealed H13 at a temperature of 650°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 13: DEFORM-2D simulations showing the effective strain rate ([mm/mm]/s) behavior—at steps 100 and 175—of annealed H13 at a temperature of 800°C and strain rates of 1s⁻¹ and 20s⁻¹.

Table 4 shows the strain rate behavior exhibited by the simulation between steps 100 and 175. It can be observed that strain rates slightly decrease with the increase in temperature; it is shown to be more significant at the center—as opposed to the border—in all cases. As the higher strain rates are employed, the effective strain rate increases.

Table 4: Summary of effective strain rate (in [mm/mm]/s) over the range of temperatures from 650°C to800°C and strain rates from 1s-1 and 20s-1 for annealed H13 samples.

Strain Ra	te erature	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
T= 🔸	Step 100	1.9	0.92	9.39	4.56	18.7	9.11	37.3	18.2
650 C	Step 175	1.45	0.68	7.3	3.5	15.5	7.05	29.4	14.2
T=	Step 100	1.65	0.91	8.19	4.56	16.3	9.11	32.5	18.2
700 C	Step 175	1.45	1.07	7.3	2.52	14.5	7.06	29.4	14.2
T=	Step 100	1.66	0.91	8.19	4.56	16.3	9.11	32.5	18.2
750 C	Step 175	1.45	1.07	7.3	3.5	14.6	7.06	29.4	14.2
T=	Step 100	1.66	0.91	8.19	4.56	16.3	9.11	32.5	18.2
800 C	Step 175	1.45	0.69	7.3	3.5	14.6	7.06	29.4	14.2

Selected excerpts from DEFORM-2D shown in *Figure 14* and *Figure 15* show the local effective strain (in [mm/mm]) at steps 100 and 175 (out of 200) of the hardened H13, at temperatures 650°C (*Figure 14*) and 800°C (*Figure 15*) and at strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 14: DEFORM-2D simulation effective strain (in [mm/mm]) results of hardened H13 at steps 100 and 175; at a temperature of 650°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 15: DEFORM-2D simulation effective strain (in [mm/mm]) results of hardened H13 at steps 100 and 175; at a temperature of 800°C and strain rates of 1s⁻¹ and 20s⁻¹.

As shown in **Table 5** below, it can be observed that as the temperature is increased, the effective strain increases—however, this trend is not observed with the higher strain rate; it can be observed that it remains similar.

Table 5: Summary of simulation results for hardened H13, showing effective strain (mm/mm) for a rangeof temperatures from 650°C to 800°C and strain rates from 1s⁻¹ to 20s⁻¹.

Strain Ra	te	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
/Temperature									
T=	Step 100	0.88	0.54	0.85	0.38	0.77	0.45	0.77	0.45
650 C	Step 175	2.35	0.9	2.2	0.85	2.19	0.7	2.17	0.7
T=	Step 100	0.7	0.54	0.74	0.4	0.77	0.45	0.77	0.45
700 C	Step 175	1.77	0.7	1.9	0.85	1.9	0.9	1.75	0.9
T=	Step 100	0.7	0.5	0.62	0.38	0.66	0.45	0.66	0.45
750 C	Step 175	1.48	0.9	1.39	0.85	1.42	0.9	1.42	0.9
T=	Step 100	1.4	0.53	0.98	0.4	0.88	0.45	0.77	0.46
800 C	Step 175	2.35	0.65	1.65	0.85	1.4	0.9	1.16	0.9

Composite images from simulations in *Figure 16* and *Figure 17* present the local effective stress (in MPa) of the hardened H13 at steps 100 and 175 (out of 200) at temperatures of 650° C and 800° C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 16: Simulation results at steps 100 and 175 (out of 200) of hardened H13 at a temperature of 650°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 17: Simulation results at steps 100 and 175 (out of 200) of hardened H13 at a temperature of 800°C and strain rates of 1S⁻¹ and 20S⁻¹.

As presented below in **Table 6**, it is evident that the effect of hardening the H13 through quenching and tempering heat treatments most significantly has an impact between the temperatures of 650°C and 700°C (where the higher stress values are exhibited). As the temperature increases—that is, within 750°C and 800°C—the decreased stress values are exhibited, as shown in the table. At steps 100 and 175, simulations performed at 750°C exhibit regular distribution of stress: with higher values observed from the center to the border. Viewing the data contained within this table, it becomes evident that as the temperature increases, there is a decrease of stress.

Table 6: Summary of hardened H13 simulation effective stress (in MPa) results over the temperaturerange of 650°C to 800°C for strain rates from 1s⁻¹ to 20s⁻¹.

Strain Rat	te	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
_ Tempe	erature								
▶ T=	Step 100	1050	1180	1100	1230	1120	1240	1140	1260
650 C	Step 175	939	1070	981	1120	1000	1130	1030	1160
T=	Step 100	764	924	842	970	862	989	883	1010
700 C	Step 175	804	804	847	847	865	865	899	899

T=	Step 100	664	534	714	586	734	608	756	629
750 C	Step 175	669	535	713	713	731	731	769	769
T=	Step 100	404	273	536	456	734	479	629	502
800 C	Step 175	400	400	444	444	597	463	640	531

The composite simulation excerpts shown below in *Figure 18* and *Figure 19* display the simulation results for hardened H13 of effective strain (in ([mm/mm]/s)) at steps 100 and 175 (out of 100), the temperatures of 650°C and 800°C, and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 18: Effective strain rate behavior (in ([mm/mm]/s)) of hardened H13 at steps 100 and 175 (out of 200) at a temperature of 650°C and with strain rates of 1s⁻¹ and 20s⁻¹.



Figure 19: Effective strain rate behavior (in ([mm/mm]/s)) of hardened H13 at steps 100 and 175 (out of 200) at a temperature of 800°C and with strain rates of 1s⁻¹ and 20s⁻¹.

The table shown below, **Table 7**, summarizes the values for strain rate obtained via modeling the hardened H13 over the range of temperatures from 650°C to 800°C (in increments of 50°). Similar to what was observed with the annealed material, it is shown that the effective strain rates slightly decrease as the temperature increases—however, note that there is a significant difference between the observed effective strain rate at the border versus the observed effective strain rate at the border. It is also evident that higher experimental strain rates resulted in higher effective strain rates.

Strain Rat	te	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
► Tempe	erature								
T=	Step 100	1.9	0.92	9.39	4.56	18.7	9.11	37.3	18.2
650 C	Step 175	1.45	0.68	7.3	3.5	15.5	7.05	29.4	14.2
T=	Step 100	1.65	0.91	8.19	4.56	16.3	9.11	32.5	18.2
700 C	Step 175	1.45	1.07	7.3	2.52	14.5	7.06	29.4	14.2
T=	Step 100	1.88	1.26	9.92	5.01	19.8	10.1	39.8	20.1
750 C	Step 175	1.58	1.06	8.27	5.72	12.4	7.38	24.8	14.8
	Step 100	3.12	0.63	9.92	5.01	15	10.1	29.8	20.1

Table 7: Summary of hardened H13 effective strain rate (in ([mm/mm]/s)) simulation results for a rangeof temperatures from 650°C to 800°C and strain rates from 1s⁻¹ to 20s⁻¹.

T=									
800 C	Step 175	1.58	1.06	5.72	3.17	12.4	7.36	24.8	14.8

The evaluation of MMC coatings to improve die life (for dies made of H13) involved testing various combinations of Inconel 625 (INC625) and Titanium Carbide (TiC). Pure INC625 was evaluated as a coating material, and INC625 in combination with various percentages of TiC mixed in by weight. Initially, INC625 with weight percentages of TiC of 5w%, 10w%, and 15w% were to be tested. However, only the INC625 with 5w% TiC produced prints of acceptable density and lack of cracks / significant pores. Despite extensive attempts at printing INC625 reinforced with 10w% TiC—including altering print parameters and mixing strategies—the resulting prints showed extensive brittleness, as well as cracking and considerable porosity.

Pure INC625 and INC625 reinforced with 5w% TiC were printed into cylindrical samples of dimensions 15mm height and 10mm diameter for use with Gleeble experiments. These samples were tested in the same manner as the annealed H13 and hardened H13: simulation results were noted at steps 100 and 175 (out of 200) for each simulation, and strain rates of 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹ were utilized, with a displacement of 10mm for each compression test. However, these simulations were run at slightly higher temperatures: the range of 750°C to 850°C was used.

Figure 20 and **Figure 21** show the composite simulation results of the local effective strain (in [mm/mm]) at steps 100 and a175 (out of 200) for the compression simulation of INC625 + 5w%TiC. The temperature parameters for these simulations are those used in the compression tests: temperatures of 750°C and 850°C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 20: Composite image showing the behavior of effective strain (in [mm/mm]) for INC625 + 5w%TiC at steps 100 and 175 (out of 200) for DEFORM-2D simulations. The temperature at which these simulations were run was 750°C; the strain rates used were 1s⁻¹ and 20s⁻¹.



Figure 21: Composite image showing the behavior of effective strain (in [mm/mm]) for INC625 + 5w%TiC at steps 100 and 175 (out of 200) for DEFORM-2D simulations. The temperature at which these simulations were run was 850°C; the strain rates used were 1s⁻¹ and 20s⁻¹.

Presented below in **Table 8** are the effective strain (in [mm/mm]) values for the INC625 + 5w%TiC simulations run at temperatures 750°C, 800°C, and 850°C and strain rates 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹. It can be observed from this table that there are no significant variations for this range of temperatures and strain rates (speeds) simulated. Examining the center region, it can be determined that an increase in temperature drives the increase in strain.

Table 8: Simulation effective strain (in [mm/mm]) results for evaluation of INC625 + 5w%TiC over temperatures 750°C, 800°C, and 850°C and strain rates 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹.

Strain Ra	te	1s ^{−1} Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
Tempe	erature								
T =	Step 100	0.75	0.46	0.75	0.46	0.72	0.45	0.7	0.44
750 C	Step 175	1.83	0.79	1.79	0.77	1.76	0.76	1.7	0.74
T=	Step 100	0.75	0.46	0.73	0.46	0.73	0.45	0.79	0.44
800 C	Step 175	1.83	0.79	1.79	0.77	1.76	0.76	1.7	0.74
T=	Step 100	0.75	0.46	0.73	0.46	0.63	0.45	0.6	0.44
850 C	Step 175	1.83	0.79	1.59	0.77	1.56	0.76	1.32	0.74

Figure 22 and **Figure 23** show the local effective stress (in MPa) at steps 100 and 175 (out of 200) for the simulation of INC625 + 5w%TiC. These simulations were run at temperatures of 750° C and 850° C, and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 22: Composite image of INC615 + 5w%TiC DEFORM-2D simulations showing the behavior of the effective stress (in MPa) for steps 100 and 175 (out of 200) at a temperature of 750°C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 23: Composite image of INC615 + 5w%TiC DEFORM-2D simulations showing the behavior of the effective stress (in MPa) for steps 100 and 175 (out of 200) at a temperature of 850°C and strain rates of $1s^{-1}$ and $20s^{-1}$.

As shown in **Table 9**, there are a few variations of effective stress (in MPa) over the specified range of temperature; this may be due to the combination of high limits of stress with this combination of materials. These results suggest that an increase in temperature would result in a decrease of effective stress observed.

Table 9: Evaluation of effective stress (in MPa) of INC615 + 5w%TiC simulations run in the temperature
range of 750°C to 850°C and at strain rates from 1s ⁻¹ to 20s ⁻¹ .

Strain Rate		1s ^{−1} Center	1s ^{−1} Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
Tempera	ture								
F T= 750	Step 100	1160	1290	1230	1290	1230	1290	1320	1320
С	Step 175	1160	1160	1170	1170	1200	1200	1230	1230
T= 800	Step 100	1160	1220	1170	1230	1180	1230	1240	1240
С	Step 175	1120	1120	1130	1130	1150	1100	1230	1230
T= 850	Step 100	1030	1030	1120	1060	1230	1120	1410	1240
C	Step 175	1030	1030	1050	1010	1150	1060	1300	1160

Shown in *Figure 24* and *Figure 25* are the local effective strain rate behavior (in [mm/mm]/s) values for simulations of INC615 + 5w%TiC at steps 100 and 175 (of 200) with applied temperatures of 750° C and 850° C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 24: Composite image showing steps 100 and 175 (out of 200) of the effective strain rate behavior for INC625 + 5w%TiC simulations, run at 750°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 25: Composite image showing steps 100 and 175 (out of 200) of the effective strain rate behavior for INC625 + 5w%TiC simulations, run at 800°C and strain rates of 1s⁻¹ and 20s⁻¹.

Shown below in **Table 10** is a summary of the effective strain (in [mm/mm]/s) for DEFORM-2D simulations of the INC615 + 5w%TiC at temperatures 750°C, 800°C, and 850°C, and strain rates $1s^{-1}$, $5s^{-1}$, $10s^{-1}$, and $20s^{-1}$. It is evident that the effective strain rises as the strain rate increases. It is also apparent that there is no substantial change in the effective strain in the range of temperatures between 750°C and 850°C (for each respective step).

Table 10: Summary of INC625 + 5w%TiC simulation effective strain (in [mm/mm]/s) results at steps 100 and 175 (out of 200) for temperatures 750°C, 800°C, and 850°C and strain rates of 1s⁻¹, 5s⁻¹, 10s⁻¹, and 20s⁻¹.

Strain Rate		► 1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
/ Temper	rature								
🕨 T= 750	Step 100	3.19	0.94	15.4	4.56	29.7	8.83	55.6	16.7
С	Step 175	1.32	0.78	6.56	3.92	13.1	7.82	25.7	15.3
T= 800	Step 100	3.19	0.9	15.4	4.56	29.7	8.83	55.6	16.7
С	Step 175	1.32	0.78	6.56	3.92	13.1	7.82	25.7	15.3
T= 850	Step 100	2.74	0.94	13.3	4.56	21.3	8.83	42	19.7
С	Step 175	1.32	0.78	6.55	3.92	13.1	7.82	25.7	15.3

Figure 26 and *Figure 27* show the local effective strain behavior (in mm/mm) at steps 100 and 175 (of 200 total) for the hot compression simulations of pure INC625. The shown simulations were performed at 750C and 850C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 26: Composite image of effective strain (in mm/mm) simulation results for pure INC625 hot compression at 750°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 27: Composite image of effective strain (in mm/mm) simulation results for pure INC625 hot compression at 850° C and strain rates of $1s^{-1}$ and $20s^{-1}$.

As presented in **Table 11** (below), for all cases, the effective strain values increase with the number of simulation steps, with higher effective strain values at the center than at the border. Virtually no variations are observed with the increase of the strain rate.

Table 11: Summary of effective strain (in mm/mm) values for pure INC625 hot compression simulations
performed in the temperature range of 750C to 850C and at strain rates of 1s ⁻¹ to 20s ⁻¹ .

Strain Rate		1s ^{−1} Center	1s ⁻¹ Border	5s ^{−1} Center	5s ^{−1} Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
Temp	erature								
► T=	Step 100	0.64	0.41	0.64	0.41	0.64	0.41	0.64	0.41
750 C	Step 175	1.75	0.79	1.74	0.79	1.74	0.79	1.74	0.79
T=	Step 100	0.64	0.41	0.64	0.41	0.64	0.41	0.64	0.41
800 C	Step 175	1.75	0.79	1.74	0.79	1.74	0.79	1.74	0.79
T=	Step 100	0.64	0.41	0.57	0.41	0.57	0.41	0.57	0.41
850 C	Step 175	1.36	0.79	1.55	0.79	1.55	0.79	1.55	0.79

Figure 28 and *Figure 29*, shown below, display the local effective stress behavior (in MPa) at steps 100 and 175 for pure INC625 at temperatures 750°C and strain rates 1s⁻¹ and 20s⁻¹.



Figure 28: Composite image showing the effective stress (in MPa) for pure INC625 hot compression simulations at steps 100 and 175, a temperature of 750°C, and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 29: Composite image showing the effective stress (in MPa) for pure INC625 hot compression simulations at steps 100 and 175, a temperature of 850°C, and strain rates of 1s⁻¹ and 20s⁻¹.

As shown below in **Table 12**, higher effective stress is observed when compared with the H13 materials, with few variations in the range of temperatures explored—suggesting an increase in temperature would lead to a decrease in stress.

Table 12: Summary of effective stress values (in MPa) for pure INC625 hot compression simulations over750°C to 850°C, at strain rates from 1s⁻¹ to 20s⁻¹.

Strain Ra	te	1s ⁻¹ Center	1s ^{−1} Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
Tempe	erature								
► T=	Step 100	1060	1190	1070	1200	1080	1200	1090	1210
750 C	Step 175	1040	1040	1060	1060	1070	1070	1050	1050
T=	Step 100	1060	1190	1070	1200	1080	1200	1090	1210
800 C	Step 175	1040	1040	1080	1030	1040	1040	1050	1050
T=	Step 100	1060	992	1070	1130	1080	1140	1090	1210
850 C	Step 175	1040	1040	1030	1030	1040	1040	1050	1050

Viewable in *Figure 30* and *Figure 31* below are the results of simulations showing the local behavior of the effective strain rate (in [mm/mm]/s) for steps 100 and 175 of pure INC625 simulations performed at 750C and 850C and strain rates of $1s^{-1}$ and $20s^{-1}$.



Figure 30: Composite image showing effective strain rate (in [mm/mm]/s) simulation results—at steps 100 and 175 (out of 200)—of pure INC625 at a temperature of 750°C and strain rates of 1s⁻¹ and 20s⁻¹.



Figure 32: Composite image showing effective strain rate (in [mm/mm]/s) simulation results—at steps 100 and 175 (out of 200)—of pure INC625 at a temperature of 850°C and strain rates of 1s⁻¹ and 20s⁻¹.

Presented in **Table 13** are the effective strain rate (in [mm/mm]/s) values resulting from the pure INC625 simulations at varying strain rates and temperatures in the range of 750°C to 850°C. There exists a considerable increase in the effective strain rate with the increase in strain rate/process speed, particularly at 20s⁻¹. However, the strain rate remains relatively unchanged in the temperature range of 750°C to 850°C.

Strain Ra	te	1s ⁻¹ Center	1s ⁻¹ Border	5s ⁻¹ Center	5s ⁻¹ Border	10s ⁻¹ Center	10s ⁻¹ Border	20s ⁻¹ Center	20s ⁻¹ Border
Temp	erature								
► T=	Step 100	2.95	0.92	14.7	4.64	29.4	9.28	58.8	18.5
750 C	Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6
T=	Step 100	2.54	0.92	12.7	4.64	21.4	9.28	50.8	18.5
800 C	Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6
T=	Step 100	2.14	0.92	10.7	4.64	21.4	9.28	50.8	18.5
850 C	Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6

Table 13: Summary of effective strain rate (in [mm/mm]/s) simulation results for INC625 in thetemperature range of 750°C to 850°C, at strain rates from 1s⁻¹ to 20s⁻¹.

Preliminary Evaluation:

This research involved the simulation of hot compression for four different materials: annealed H13, hardened H13, INC625 + 5w%TiC, and pure INC625. It is essential to be aware of the plastic deformation behavior of components involved in the metal-forming process from these results and use this data to address forging die concerns; for the specific use case this research addresses, the involved consideration/application is for cylindrical parts. It is important to consider homogeneous conformation conditions and then evaluate a range of deformations which make use of this specific geometry (for dies or molds). The yield stress of uniaxial compression tests as a function of strain, strain rate, and temperature, can produce the necessary data for flow stress comparisons. The hardness which results from specific heat treatments can bring about the necessary, suitable properties required for parts needing to undergo certain amounts of compression, exhibit hardness in higher temperatures, and possess fatigue- and wear-resistance.

A comparison of the simulation results for annealed H13 and hardened H13 can be made with data from published literature; it shows that when the steel receives the hardening heat treatments (quenching and tempering), there is a greater resistance to stress at higher temperatures (confirmed in the Gleeble compression tests and simulation process). H13 steel is a type of medium-carbon, hot work steel which is used to make tools for cutting, forming, and shaping materials—therefore, this steel must exhibit higher hardness for resistance to compression as well as exhibit higher hardness in higher temperature.

There exists strong potential for additive manufacturing to form the coatings which contain the necessary wear properties for dies. The results observed in this report—as well as data encountered during the literature review—support the idea that the LBPF-printed pure INC625 as well as INC625 + 5w%TiC are contenders for effective H13 die coatings. This research focused on evaluation of strain, strain rate, and stress with regards to hardened H13, which showed better resistance when compared with higher stress and high temperatures

The model presented below in *Figure 32*, *Figure 33*, and *Figure 34* shows the early stages of a simulation examining the boundary between the INC625 + 5w%TiC for coating and hardened H13. The following steps would involve evaluating the behavior of this material with the goal of assessing its response to wear on the superficial parts of hardened H13 dies.



Figure 32: Simulation prepared with hardened H13 and INC625 interface + 5w%TiC at steps 80 and 120 (of 200 total) to observe the effective strain (in mm/mm).



Figure 33: Simulation prepared with the interface of hardened H13 and INC625 + 5w%TiC at steps 80 and 120 (of 200 total) to observe the effective strain rate (in [mm/mm]/s).



Figure 34: Simulation prepared with the interface of hardened H13 and INC625 + 5w%TiC at steps 80 and 120 (of 200 total) to observe the effective stress (in MPa).

The composite image, *Figure 35*, shows a proposed application for the results obtained from the research conducted in this report. The model for a forging process is shown below, using hardened H13 as the die and punch material and examining the effect on wear behavior, where the tools to be coated with INC625 + 5w%TiC.



Figure 35: Model for hot forging process prepared for wear-coating evaluation in DEFORM-2D and DEFORM-3D.

2. Identify any project changes, and justification:

- Elimination of INC625 + 10w%TiC and INC625 + 15w%TiC from consideration
- Assessment of pure INC625

Summary of above points:

Several parameter changes and setup considerations were explored to produce INC625 + 10w%TiC coupons with less porosity / better densification and lack of cracks. However, the INC625 + 10w%TiC coupons consistently exhibited a high degree of porosity and cracking; the overall brittleness of the printed coupons led to some of the coupons shearing off of the substrate when the powder spreading wiper made contact with them. As a result, focus shifted to optimizing densification of the INC625 + 5w%TiC coupons.

INC625 + 15w%TiC has not been explored due to the issues already surrounding the INC625 + 10w%TiC coupon density. Additionally, as this research builds off of Loewy Institute PhD candidate Trevor Verdonik's work (which identified INC625 as a strong contender for forging tooling coating from an assortment of other candidates), it was deemed necessary to perform separate Gleeble experiments with pure INC625 in order for the results to be compared with the MMCs.

3. Project still on track?

- Performing additional MMC Gleeble experiments
- Continued revision of simulations in DEFORM-2D

Summary of above points:

As additional samples of INC625 + 5w%TiC are printed with the Renishaw AM400, additional Gleeble experiments will be run to further gather stress-strain curves to inform additional DEFORM-2D modeling.

4. Plans for next stage (or to complete):

- Additional experimental investigation into printing INC625 + 5w%TiC coating atop industry partner-provided H13 knockout (K.O.) pins
- Metallographic assessment of K.O. pin coatings
- Collaboration with industry partner, AAM, to assess coating effectiveness in industrial environment

Summary of above points:

An initial print of the INC625 + 5w% coating atop 1 of the 3 industry partner-provided K.O. pins resulted in poor densification, potentially due to the powder packing method used. To address this, it

was decided that several smaller hardened H13 samples should be printed upon to further assess how the packing technique may affect the final print. Additionally, it has been discussed that some initial printing of pure INC625 on the first few layers of the hardened H13 (prior to print the INC625 + 5w%TiC) may assist the remaining portion of the coating—comprised of INC625 + 5w%--to adhere to the K.O. pin.

Once the coating achieves the desirable densification, metallography will be performed on the coating, and another pin will be coated and sent to AAM for active use in one of their production lines. After the K.O. pin has completed a set lifecycle, it will be removed from production and the surface will be assessed.

5. Any concerns about completing the planned work?

• Currently, there are no concerns regarding the completion of the planned work.

Summary of the above point(s):

At this point there are no concerns regarding completion of the planned work. A methodology has been developed: from print evaluation / metallography, to Gleeble testing, to DEFORM modeling, which will be used to complete the scope of this stage gate of work.

6. Attach an updated milestone plan.

• This project is completed at this point.

APPENDICES OVERVIEW

APPENDIX A: References

APPENDIX B: Additional DEFORM Simulation Images

APPENDIX A: References (Organized by Project Topics)

Topic: DEFORM

Khanawapee, Uten and Butdee, Suthep. "A study of barreling and DEFORM 3D simulation in cold upsetting of bi-material." *MaterialsToday: PROCEEDINGS*, Vol. 26, Pt. 2, 2020, pp. 1262-1270.

Seriacopi V., et al. "Finite element analysis of the effects of thermos-mechanical loadings on a tool steel microstructure." *Engineering Failure Analysis*, Vol. 98, 2019, pp. 383-398.

Rajendran, Nijenthan et al. "Hot Forging Die Design Optimization Using FEM Analysis for Near-Net Forming of 18CrNiMo7-6 Steel Pinion Shaft." *Metals*, Vol. 13, No. 4, 2023.

Topic: H13 Surface Evaluation

Castro, G. et al. "Influence of the nitriding time in the wear behaviour of an AISI H13 steel during a crankshaft forging process." *Wear*, Vol. 263, Iss. 7-12, pp. 1275-1385.

Chen, Yong-gui et al. "Improvement of the wear and corrosion resistance of nitrocarburized H13 steel using hydrothermal-synthesized zeolite coating." *Surface and Coatings Technology*, Vol. 447, 2022.

Pérez, Marcos, and Belzunce, Francisco Javier. "A comparative study of salt-bath nitrocarburizing and gas nitriding followed by post-oxidation used as surface treatments of H13 hot forging dies."

da Silva, Sinval Pedroso, et al. "Surface modification of AISI H13 steel by die-sinking electrical discharge machining and TiAIN coating: A promising hybrid technique to improve wear resistance." *Wear*, Vol. 462-463, 2020.

Wang, Yanjiang, et al. "Determining the wear behavior of H13 steel die during the extrusion process of pure nickel." *Engineering Failure Analysis*, Vol. 134, 2022.

Topic: TiC Reinforcement

Cho, Seungchan et al. "Enhanced high-temperature compressive strength of TiC reinforced stainless steel matrix composites fabricated by liquid pressing infiltration process." *Journal of Alloys and Compounds*, Vol. 817, 2020.

Yang, Xiaoyu et al. "Effect of carbon content on interfacial microstructure and mechanical properties of a vacuum hot-compressed bonding titanium-steel composite." *Materials Science and Engineering: A*, Vol. 824, 2021.

Comparisons and evaluations through Strain Rates, Strain and Stress Materials: H13 annealed, H13 Hardened, INC625 + 5w%TiC, Pure INC625 and H13 HARDENED + (INC625 + 5w%TiC)											
		ession									
Sample	STRAIN RATE	650°C	700°C	750°C	800°C	825°C	850°C				
H15 mm x D10 mm	1s-1, 5s-1, 10s-1, and 20s-1										
Single	H13 ANNEALED	Х	Х	Х	Х						
Single		x	x	x	x						
5Bic		X	A	~	X						
Single	INC625 + 5w%TiC			х	х	х	х				
Single	Pure INC625			х	х	х	х				
Multiple	H13 HARDENED + (INC625 + 5w%TiC)			х	х						

APPENDIX B: Additional DEFORM Simulation Images








































88.2

Min 88.2 Max 245

800C,

1s⁻¹

Min

800C,

5s-1



Material: H13 annealed					Strain Rate – Effective														
Steps of the simulations 100				00	((mm/mm)/s)														
of 200 and 175 of 200					T= 650 C	1 S ⁻¹ Center	1 S ^{−1} Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻ Borde	-1 er					
					Step 100	1.9	0.92	9.39	4.56	18.7	9.11	37.3	18.2	2					
					Step 175	1.45	0.68	7.3	3.5	15.5	7.05	29.4	14.2	2					
					T= 700 C														
					Step 100	1.65	0.91	8.19	4.56	16.3	9.11	32.5	18.2	2					
					Step 175	1.45	1.07	7.3	2.52	14.5	7.06	29.4	14.2	2					
					T= 750 C														
					Step 100	1.66	0.91	8.19	4.56	16.3	9.11	32.5	18.2	2					
					Step 175	1.45	1.07	7.3	3.5	14.6	7.06	29.4	14.2	2					
					T= 800 C														
Strain – Effective (mm/mm))	Step 100	1.66	0.91	8.19	4.56	16.3 9.11		32.5	32.5 18.2			Stress – Effective (MPa			MPa)	
		•			Step 175	1.45	0.69	7.3	3.5	14.6	7.06	29.4	14.2	2					
T= 650 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border		T= 6	50 C 1 Ce	S ⁻¹ 1 nter Bo	L S ⁻¹ order	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border
Step 100	0.6	0.45	0.59	0.45	0.59	0.45	0.59	0.45		Step :	100 2	22 :	183	302	252	345	289	395	332
Step 175	1.33	0.77	1.33	0.77	1.33	0.77	1.33	0.77		Step 3	175 2	.05	185	283	258	325	297	373	311
T= 700 C										T= 7	'00 C								
Step 100	0.6	0.45	0.59	0.45	0.59	0.45	0.59	0.45		Step :	100 1	.83 :	154	277	227	317	251	383	301
Step 175	1.33	0.77	1.33	0.77	1.33	0.77	1.33	0.77		Step 3	175 1	.85 :	157	258	233	297	269	342	311
T= 750 C										T= 7	'50 C								
Step 100	0.6	0.46	0.66	0.45	0.59	0.45	0.59	0.45		Step :	100 1	.64	144	227	201	261	233	332	270
Step 175	1.19	0.77	1.33	0.77	1.33	0.77	1.33	0.77		Step 3	175 1	.47	127	233	183	269	214	311	250
T= 800 C										T= 8	00 C								
Step 100	0.53	0.46	0.66	0.45	0.52	0.45	0.59	0.45		Step :	100 1	.44 :	124	201	176	233	205	301	238
Step 175	1.19	0.77	1.19	0.77	1.19	0.77	1.19	0.77		Step :	175 1	.27 :	108	183	158	241	214	281	220



2.06

1.77

1.19

0.903

0.613

0.324

0.0341 Min 0.0341 Max 2.35

650C,

1s-1

1.93

1.66

1.39

1.13

0.857

0.589

0 321

Min 0.0523 Max 2.20

650C,

5s-1































Strain Rate – Effective																		
Material: H13 Hardened								((m	m/mm)/s)								
Steps of the simulations 100 of 200 and 175 of 200					T= 650 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border					
					Step 100	1.9	0.92	9.39	4.56	18.7	9.11	37.3	18.2					
					Step 175 T= 700 C	1.45	0.68	7.3	3.5	15.5	7.05	29.4	14.2					
					Step 100	1.65	0.91	8.19	4.56	16.3	9.11	32.5	18.2					
					Step 175	1.45	1.07	7.3	2.52	14.5	7.06	29.4	14.2					
					T= 750 C													
					Step 100	1.88	1.26	9.92	5.01	19.8	10.1	39.8	20.1					
					Step 175	1.58	1.06	8.27	5.72	12.4	7.38	24.8	14.8					
					1= 800 C													
Strain – Effective (mm/mm)				n)	Step 100	3.12	0.63	9.92	5.01	1 15 10.1		29.8 20.1			Strog	s – Effa	activa (MPa)
			,	-7	Step 175	1.58	1.06	5.72	3.17	12.4	7.36	24.8	14.8		Juca		.cuve (ivii aj
T= 650 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border		T= 65	50 C 1 S Cen	ter Bo	S ⁻¹ 5S rder Cen	-1 5 S ⁻¹ er Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border
Step 100	0.88	0.54	0.85	0.38	0.77	0.45	0.77	0.45		Step 1	00 10	50 11	.80 110	0 1230	1120	1240	1140	1260
Step 175	2.35	0.9	2.2	0.85	2.19	0.7	2.17	0.7		Step 1	75 93	9 10	70 98	1 1120	1000	1130	1030	1160
T= 700 C										T= 70	00 C							
Step 100	0.7	0.54	0.74	0.4	0.77	0.45	0.77	0.45		Step 1	00 76	i4 9	24 84	2 970	862	989	883	1010
Step 175	1.77	0.7	1.9	0.85	1.9	0.9	1.75	0.9		Step 1	75 80	4 8	04 84	7 847	865	865	899	899
T= 750 C										T= 75	50 C							
Step 100	0.7	0.5	0.62	0.38	0.66	0.45	0.66	0.45		Step 1	00 66	i4 5	34 71	4 586	734	608	756	629
Step 175	1.48	0.9	1.39	0.85	1.42	0.9	1.42	0.9		Step 1	75 66	95	35 71	3 713	731	731	769	769
T= 800 C										T= 80	00 C							
Step 100	1.4	0.53	0.98	0.4	0.88	0.45	0.77	0.46		Step 1	00 40	4 2	73 53	5 456	734	479	629	502
Step 175	2.35	0.65	1.65	0.85	1.4	0.9	1.16	0.9		Step 1	75 40	0 4	00 44	444	597	463	640	531




























Mate Steps of	erial: IN of the 200 an	C625 + simula d 175 d	5w%T tions 1 of 200	iC 00				Strain	Rate –	Effectiv								
								((n	nm/mr	n)/s)								
					T= 750 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border					
					Step 100	3.19	0.94	15.4	4.56	29.7	8.83	55.6	16.7					
					Step 175	1.32	0.78	6.56	3.92	13.1	7.82	25.7	15.3					
					T= 800 C													
					Step 100	3.19	0.9	15.4	4.56	29.7	8.83	55.6	16.7					
					Step 175	1.32	0.78	6.56	3.92	13.1	7.82	25.7	15.3					
					T= 850 C													
					Step 100	2.74	0.94	13.3	4.56	21.3	8.83	42	19.7					
					Step 175	1.32	0.78	6.55	3.92	13.1	7.82	25.7	15.3					
Strain	– Effec	tive (m	nm/mn	ר)											Stres	s – Effe	ective	(MPa)
T= 750 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border		T= 750 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border
Step 100	0.75	0.46	0.75	0.46	0.72	0.45	0.7	0.44		Step 100	1160	1290	1230	1290	1230	1290	1320	1320
Step 175	1.83	0.79	1.79	0.77	1.76	0.76	1.7	0.74		Step 175	1160	1160	1170	1170	1200	1200	1230	1230
T= 800 C										T= 800 C								
Step 100	0.75	0.46	0.73	0.46	0.73	0.45	0.79	0.44		Step 100	1160	1220	1170	1230	1180	1230	1240	1240
Step 175	1.83	0.79	1.79	0.77	1.76	0.76	1.7	0.74		Step 175	1120	1120	1130	1130	1150	1100	1230	1230
T= 850 C										T= 850 C								
Step 100	0.75	0.46	0.73	0.46	0.63	0.45	0.6	0.44		Step 100	1030	1030	1120	1060	1230	1120	1410	1240
Step 175	1.83	0.79	1.59	0.77	1.56	0.76	1.32	0.74		Step 175	1030	1030	1050	1010	1150	1060	1300	1160



0.600

0.409

0.218 Min 0.218 Max 1.75 0.599

0.408

Min 0.217 Max 1.74

























Ma Steps of	aterial: of the 200 an	Pure II simula d 175 d	NC625 tions 1 of 200	.00				Strain I	Rate –	Effectiv	e							
								((n	nm/mr	n)/s)								
					T= 750 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border					
					Step 100	2.95	0.92	14.7	4.64	29.4	9.28	58.8	18.5					
					Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6					
					T= 800 C													
					Step 100	2.54	0.92	12.7	4.64	21.4	9.28	50.8	18.5					
					Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6					
					T= 850 C													
					Step 100	2.14	0.92	10.7	4.64	21.4	9.28	50.8	18.5					
C 1	F ((),				Step 175	1.42	0.87	6.69	4.15	12.9	8.06	33.8	15.6		C 1	F (C)		
Strain	– Епес	tive (m	ım/mn	n)											Stress	– Епе	ctive (I	vipa)
T= 750 C	1 S⁻¹ Center	1 S⁻¹ Border	5 S⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border		T= 750 C	1 S ⁻¹ Center	1 S ⁻¹ Border	5 S ⁻¹ Center	5 S ⁻¹ Border	10 S ⁻¹ Center	10 S ⁻¹ Border	20 S ⁻¹ Center	20 S ⁻¹ Border
Step 100	0.64	0.41	0.64	0.41	0.64	0.41	0.64	0.41		Step 100	1060	1190	1070	1200	1080	1200	1090	1210
Step 175	1.75	0.79	1.74	0.79	1.74	0.79	1.74	0.79		Step 175	1040	1040	1060	1060	1070	1070	1050	1050
T= 800 C										T= 800 C								
Step 100	0.64	0.41	0.64	0.41	0.64	0.41	0.64	0.41		Step 100	1060	1190	1070	1200	1080	1200	1090	1210
Step 175	1.75	0.79	1.74	0.79	1.74	0.79	1.74	0.79		Step 175	1040	1040	1080	1030	1040	1040	1050	1050
T= 850 C										T= 850 C								
Step 100	0.64	0.41	0.57	0.41	0.57	0.41	0.57	0.41		Step 100	1060	992	1070	1130	1080	1140	1090	1210
Step 175	1.36	0.79	1.55	0.79	1.55	0.79	1.55	0.79		Step 175	1040	1040	1030	1030	1040	1040	1050	1050





















Updates of compression between two materials with the objective to evaluate the behavior, mainly on the boundary.

Comparison between the behavior of INC625 and IN625 + 5w%TiC.















